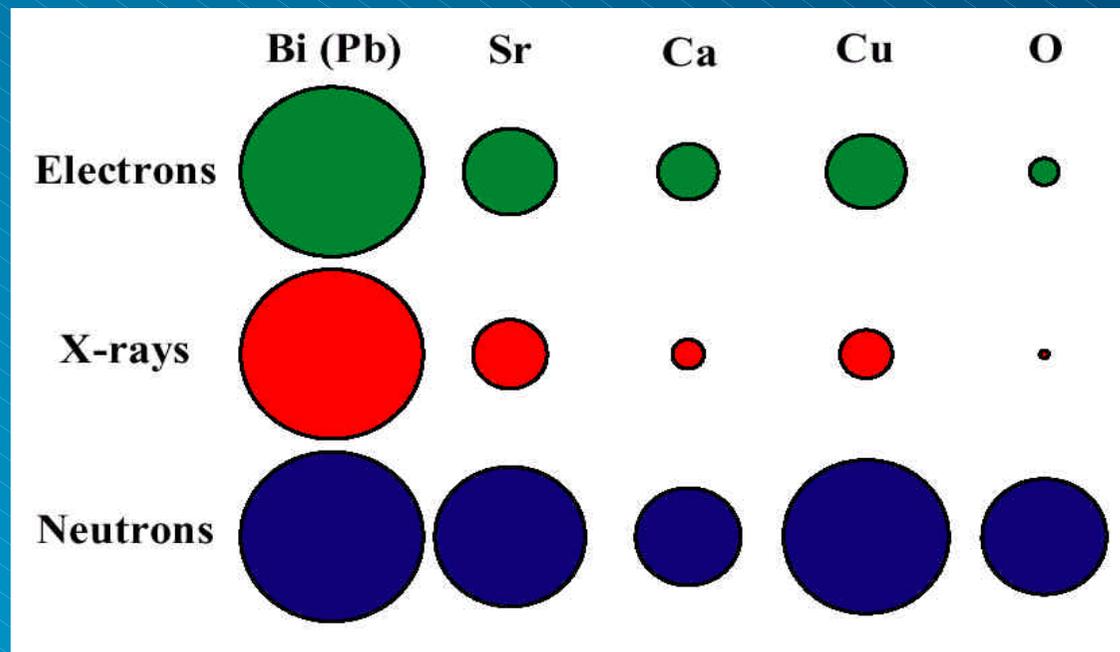




# Why Neutrons ?

- Relative Scattering Powers of the Elements

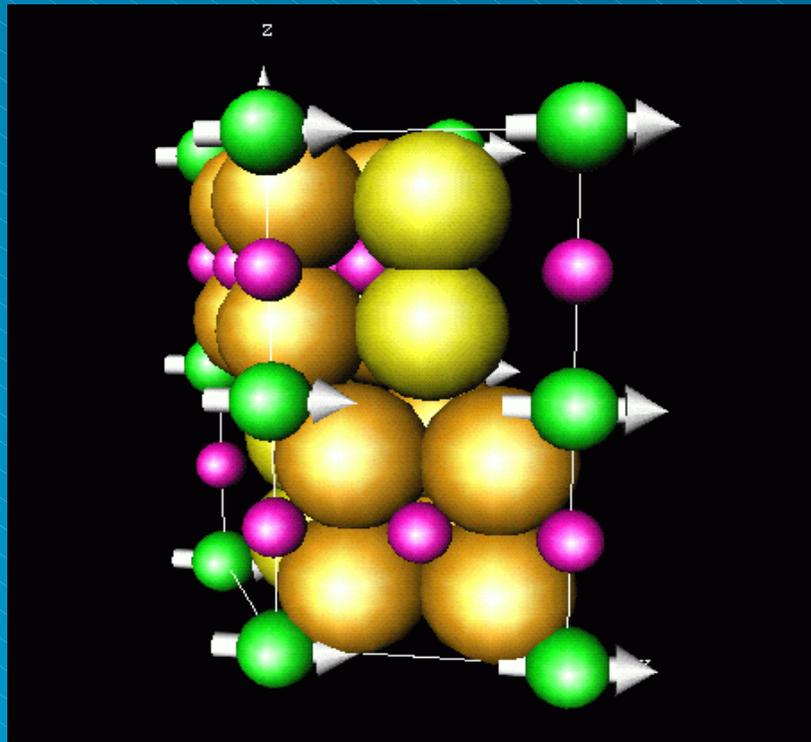


- Neutrons scatter strongly from light elements  
(Because neutron scattering is a nuclear interaction)



# Why Neutrons ?

- Neutrons are unique for Magnetic Structures



- H.M. Rietveld

## Structure of Magnetic Materials

MnTa<sub>4</sub>S<sub>8</sub> - the famous example given in the original Rietveld manual



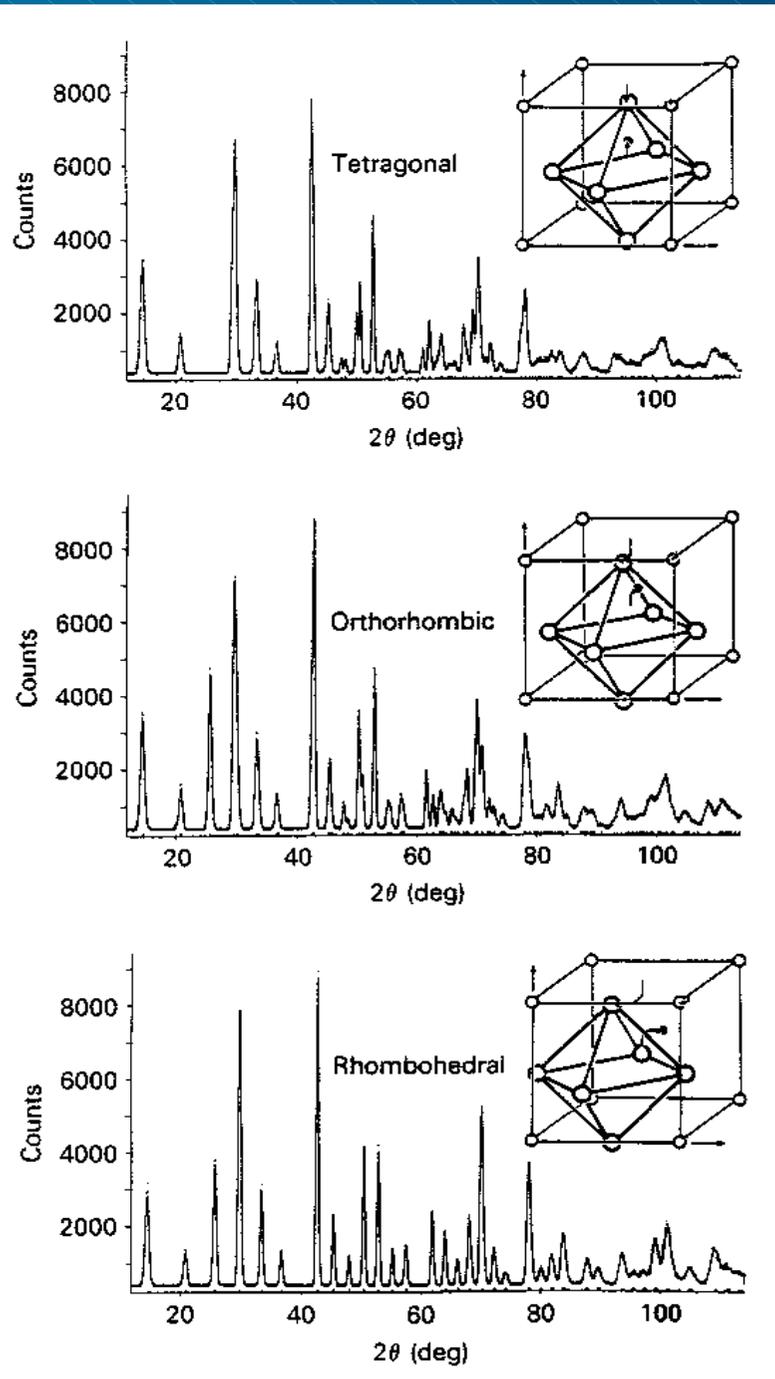
# Why Powders ?

- ...Well, if you don't have a single crystal...
- For many new, interesting materials, single crystals are not available
  - Zeolites, Superconductors, GMR materials...
- And many other materials are not really single crystals
  - At least not at 0 K, the most important temperature



# Why Powders ?

- Destructive Phase T/Ns
  - Classical Perovskite transitions  
Small displacements of light atoms
  - Subtle changes in the powder 'profile'  
- interest of "Profile Refinement"
- And no single crystals
- Examples:
  - [KNbO<sub>3</sub>](#)
  - [NaNbO<sub>3</sub>](#)





# Why Rietveld Refinement ?

- Strongly overlapping reflections
  - Previously, integrated intensities were obtained for groups of overlapping reflections.
- Key to success of RR
  - inclusion of all the information
  - refinement of physically meaningful parameters  
(reduction of correlation between parameters)



# Why not X-ray Powder Diffraction ?

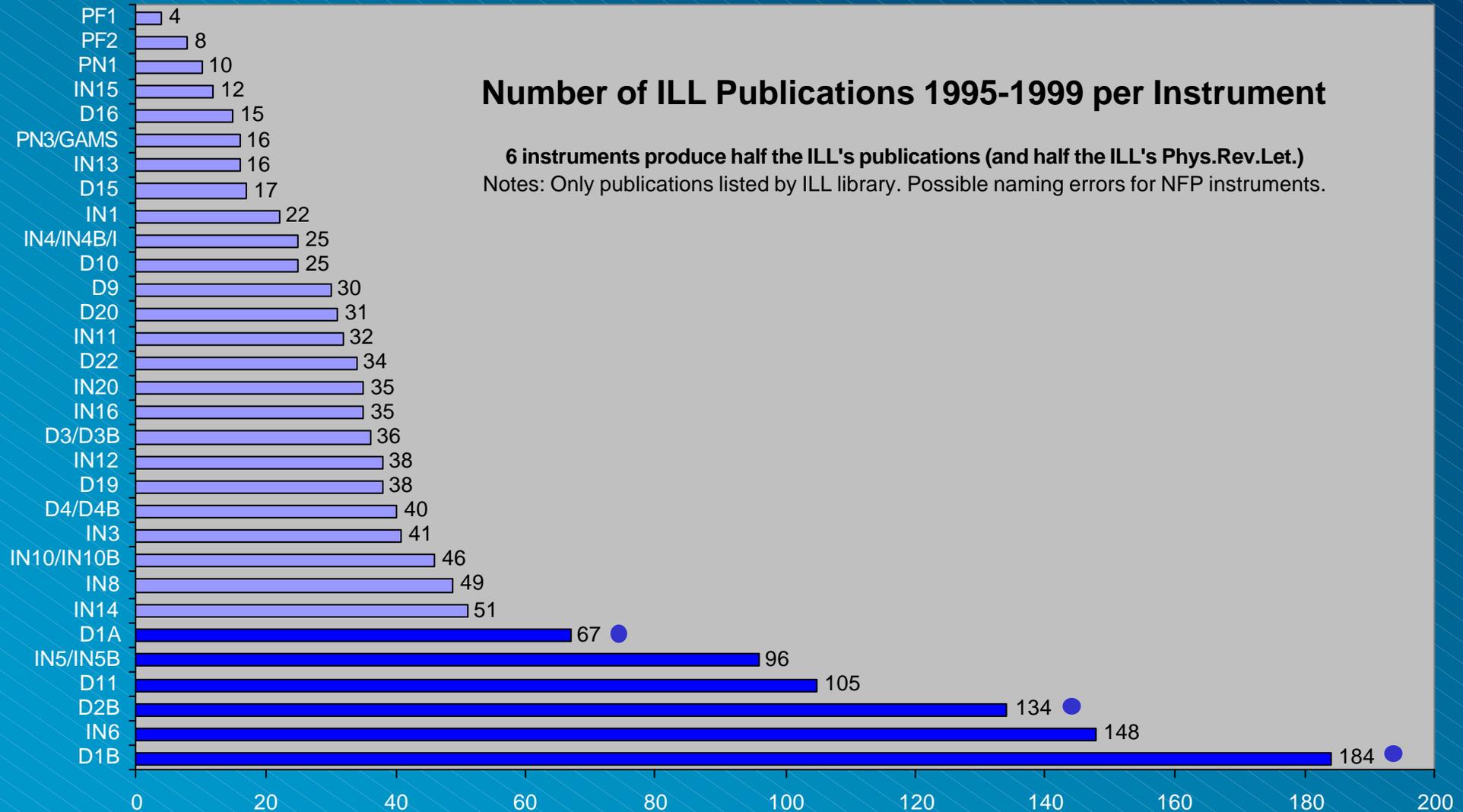
- Magnetic structures... not possible with x-ray powders
- X-rays best (synchrotrons) for **SOLVING** structures
  - Easier to find the heavy atoms first
  - All atoms are 'equal' for neutrons
- Neutrons are best for **REFINING** structures
  - Few systematic errors (average over big samples etc...)
  - Easier sample environment (low temperatures etc...)
- Interest of very precise structure measurements
  - Precise bond lengths
  - Study charge ordering, metal-insulator transitions...



# Popularity of Neutron Powder Diffraction

## Number of ILL Publications 1995-1999 per Instrument

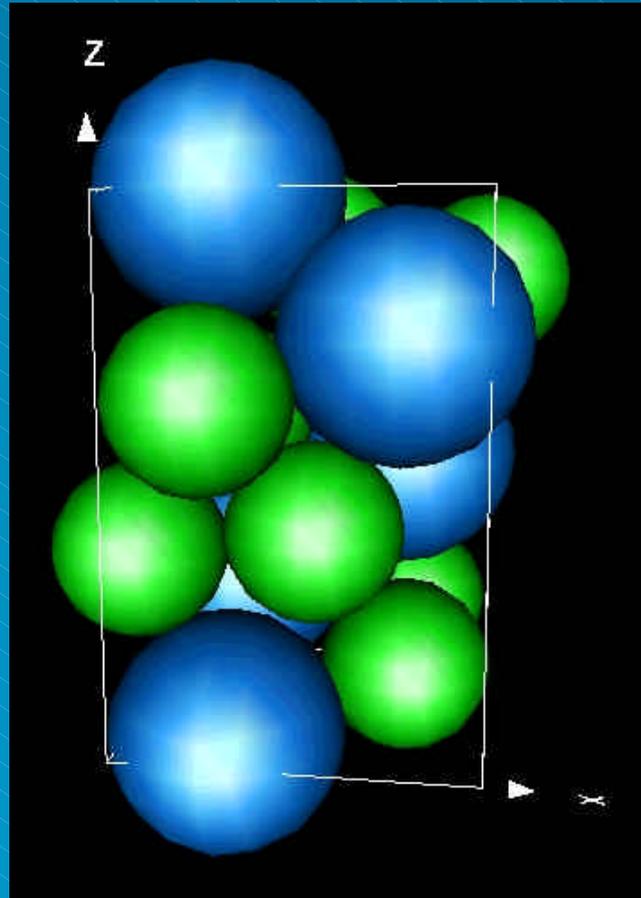
**6 instruments produce half the ILL's publications (and half the ILL's Phys.Rev.Let.)**  
 Notes: Only publications listed by ILL library. Possible naming errors for NFP instruments.





# Neutron Powder Diffraction

Real Materials, not crystals - Hydrogen in Metals



- Hydrogen storage in metals
  - Location of H among heavy atoms
  - No single crystals
  
- Laves phases eg  $\text{LnMg}_2\text{H}_7$  (La, Ce)
  - Binary alloys with large/small atoms
  - Various stackings of tetrahedral sites -can be occupied by H-atoms
  - Up to 7 Hydrogens per unit
  
- Can even find H in Eu on D20 !

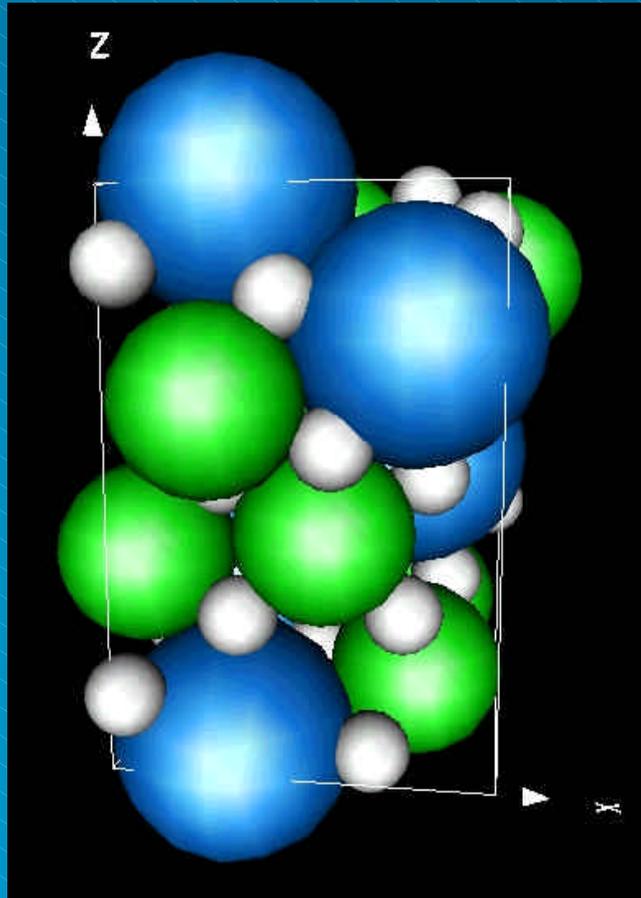
Gingl, Yvon et al. (1997) *J. Alloys Compounds* **253**, 313.

Kohlmann, Gingl, Hansen, Yvon (1999) *Angew. Chemie* **38**, 2029. etc..



# Neutron Powder Diffraction

Real Materials, not crystals - Hydrogen in Metals



- Hydrogen storage in metals
  - Location of H among heavy atoms
  - No single crystals
  
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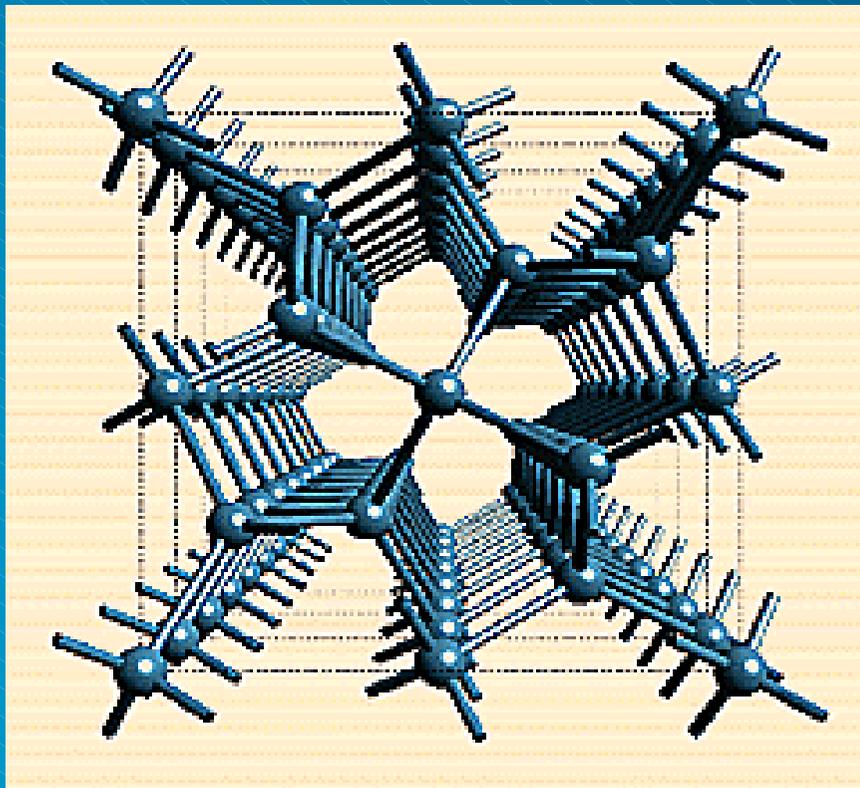
Gingl, Yvon et al. (1997) *J. Alloys Compounds* **253**, 313.

Kohlmann, Gingl, Hansen, Yvon (1999) *Angew. Chemie* **38**, 2029. etc..



# High Pressure Powder Diffraction

New phases of Ice discovered by neutron diffraction



- Mixture of 5- and 7-membered rings of Ice XII.
- Delicate balance between competing ice phases - tests water potential functions in chemical & biological systems
- Model metastable structures

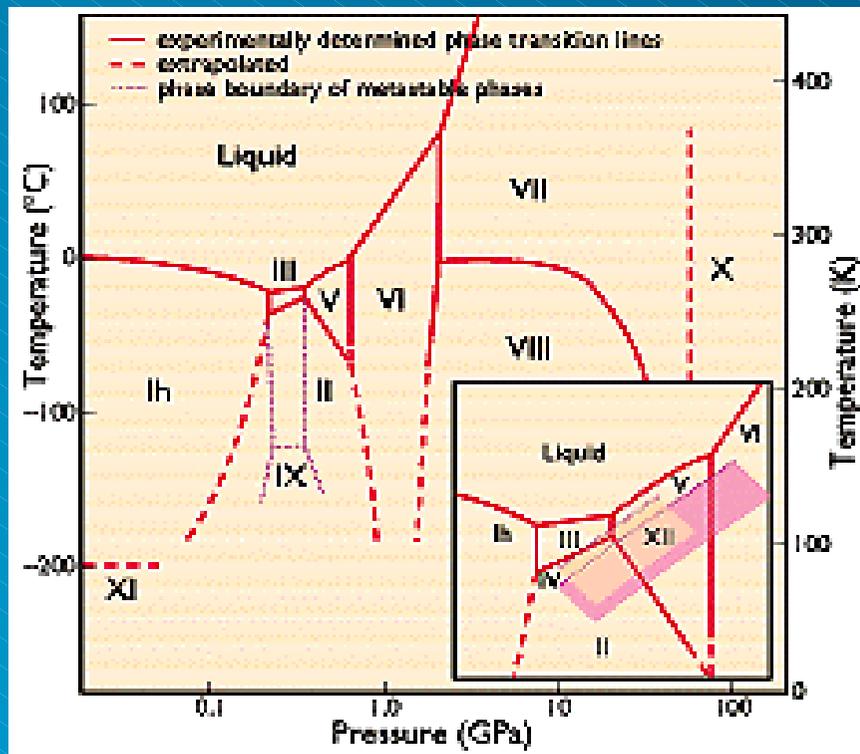
Lobban, Finney, Kuhs (1998) Nature 391, 268.

Kuhs, Lobban, Finney (1999) Rev.High Press.Sci.& Tech. 7.



# High Pressure Powder Diffraction

New phases of Ice discovered by neutron diffraction



- Ice-XII - densest form of ice without interpenetration
- Ice-IV - auto-clathrate interpenetration of H-bonds for even higher density
- Ice-He clathrate like Ice-II

Lobban, Finney, Kuhs (1998) Nature 391, 268.

Kuhs, Lobban, Finney (1999) Rev.High Press.Sci.& Tech. 7.

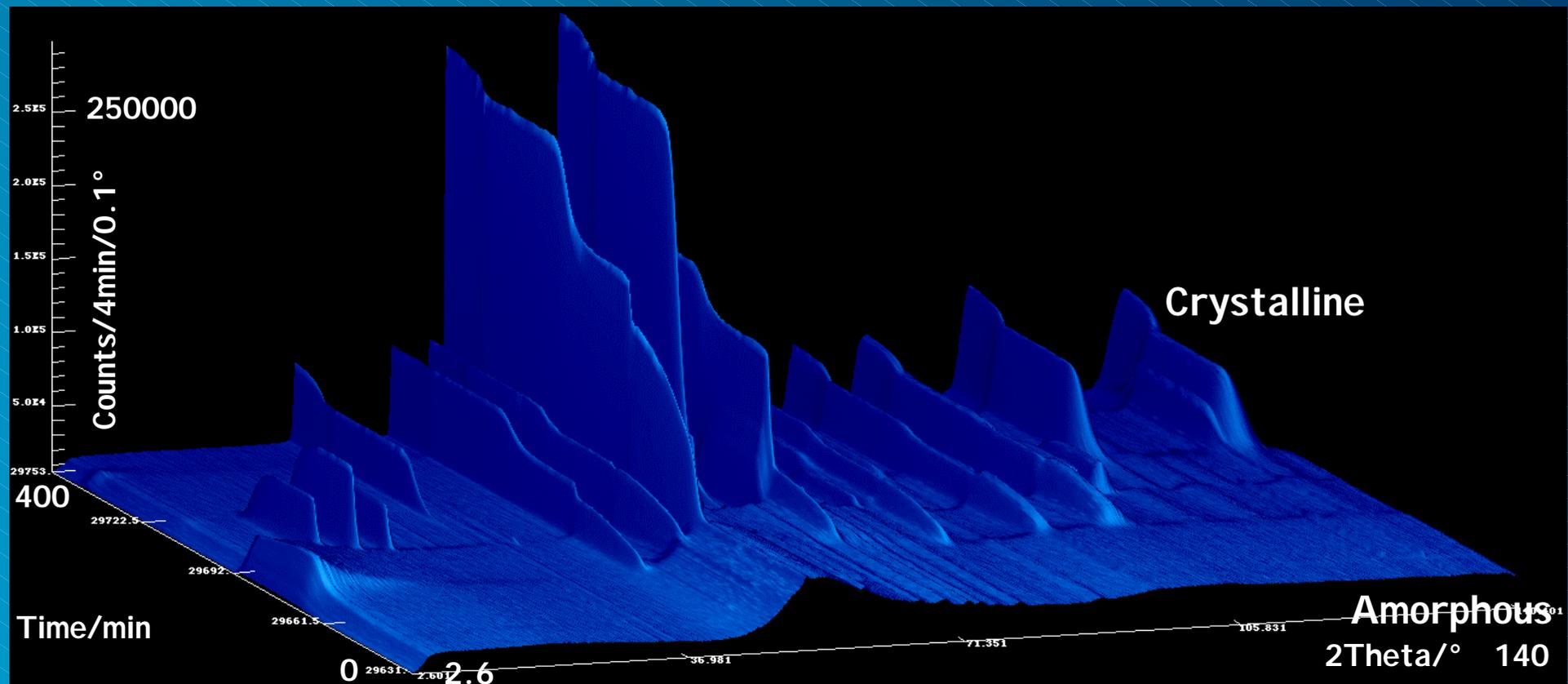


# Applications of large fast detectors

## Real-time Phase Diagrams

Sue Kilcoyne, Bob Cywinski et al.

Crystallisation of amorphous alloys  $Y_{67}Fe_{33}$  with increasing temperature



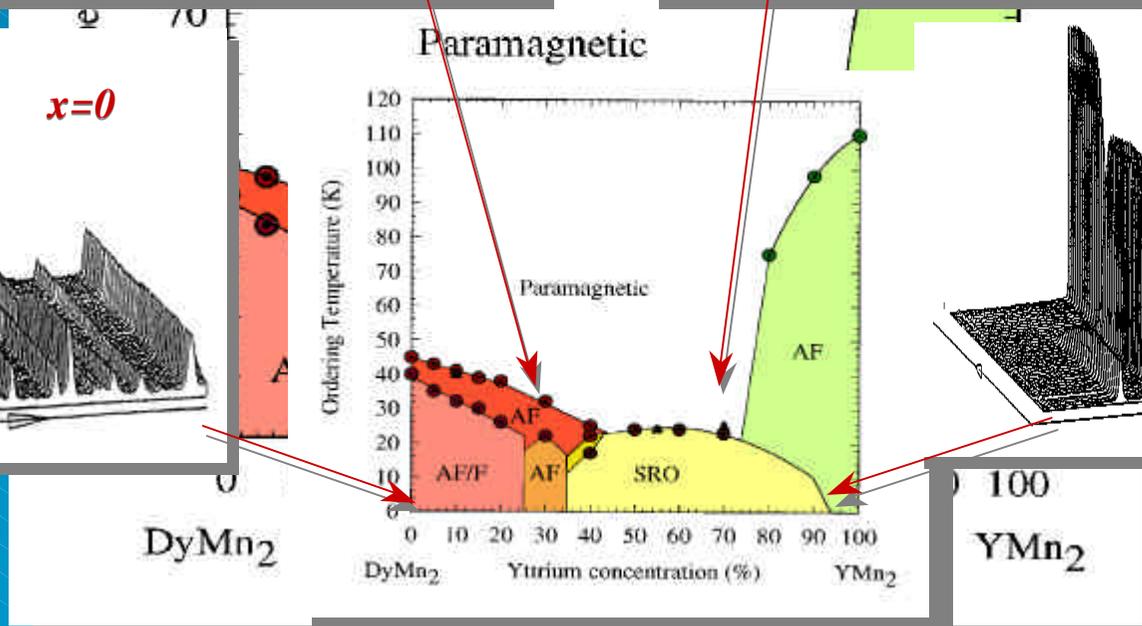
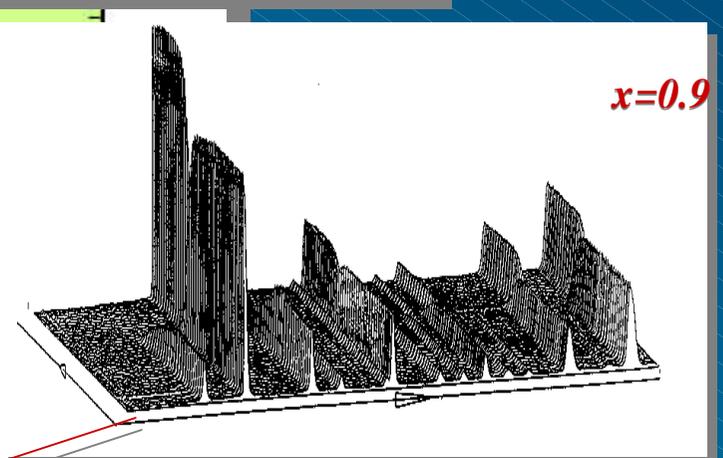
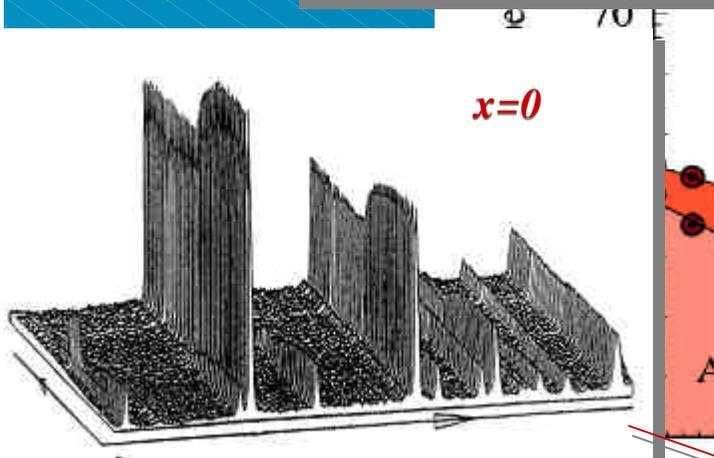
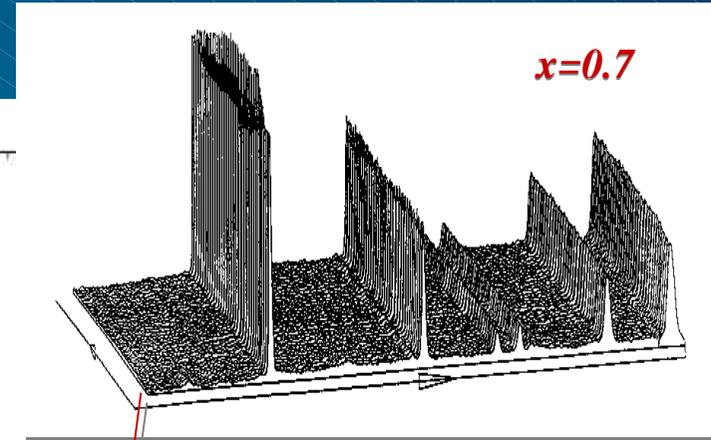
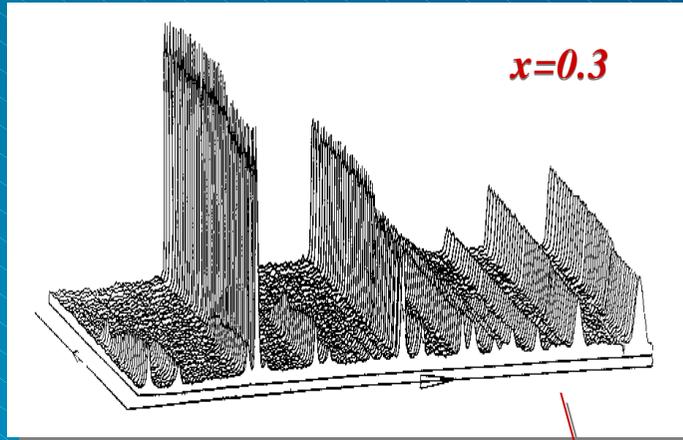
Complete diffraction pattern in minutes or seconds, scan through temperature



# Applications of large fast detectors

Pseudo-binary  $\text{RMn}_2$  compounds:  $\text{Dy}_{1-x}\text{Y}_x\text{Mn}_2$

Clemens Ritter, R. Cywinski et al on D1B

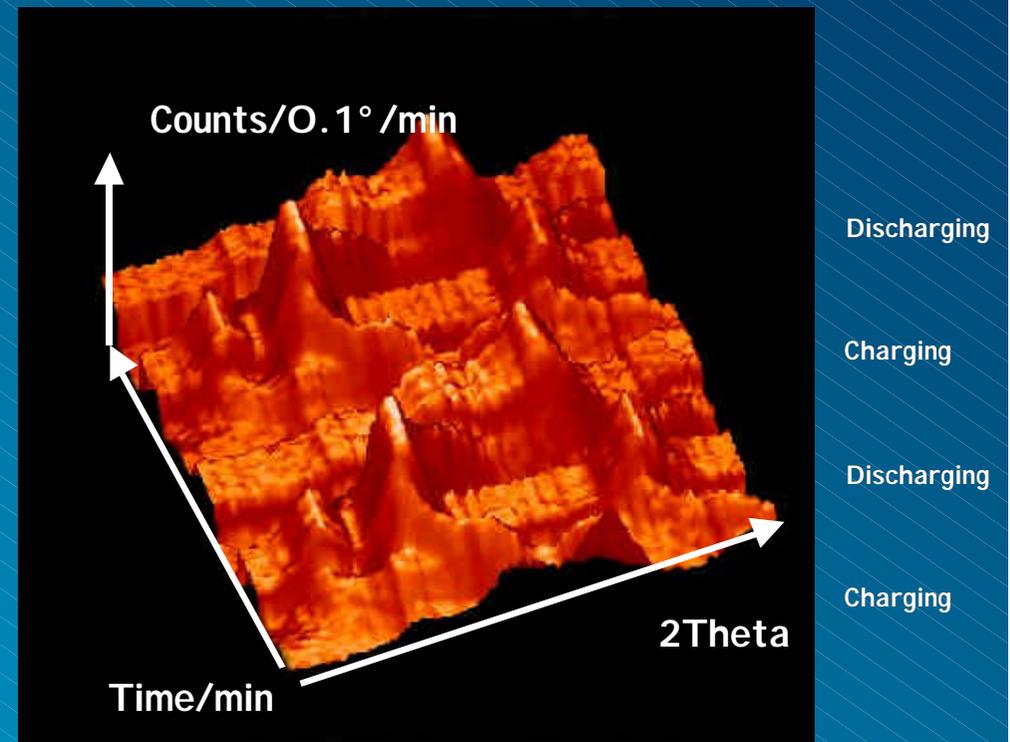
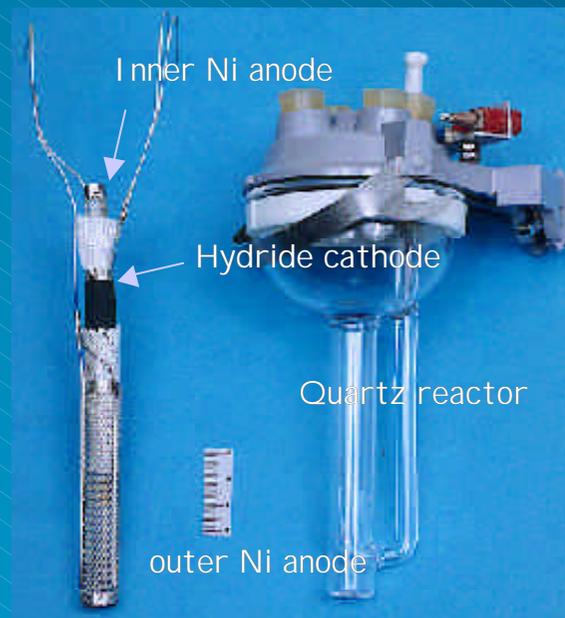




# Applications of large fast detectors

## Real-time electro-chemistry

- Latroche, Chabre et al.  
In-situ Charging and discharging of metal hydride electrodes LaNi<sub>5</sub>



- Follow chemical changes with battery charge/discharge cycle

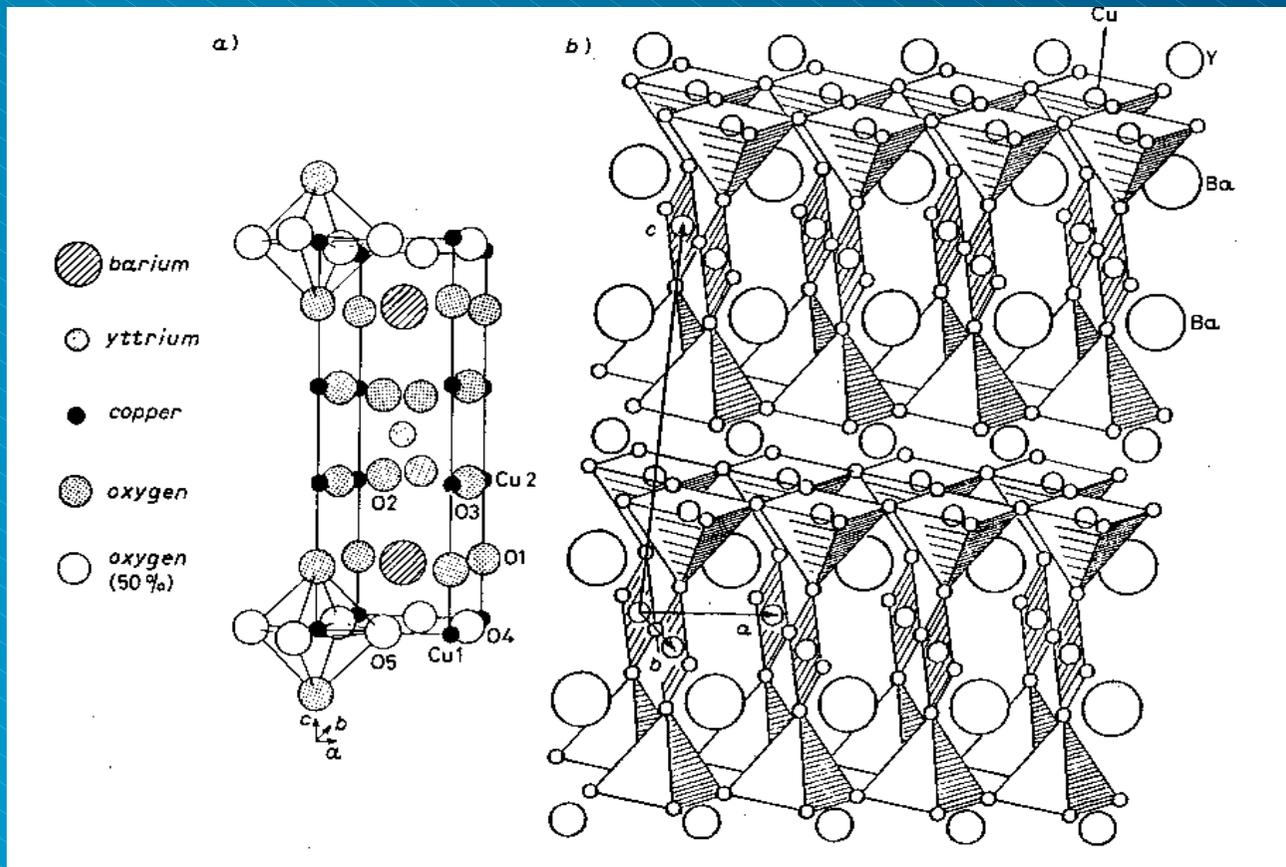
# Neutron Powder Diffraction and Novel Materials

Alan Hewat



ILL Grenoble

## Why Use Neutron Powder Diffraction ?



- Structure of the 90K high Tc superconductor
  - Left -by X-rays (Bell labs & others)
  - Right -by Neutrons (many neutron labs)
- The neutron picture gave a very different idea of the structure - important in the search for similar materials.

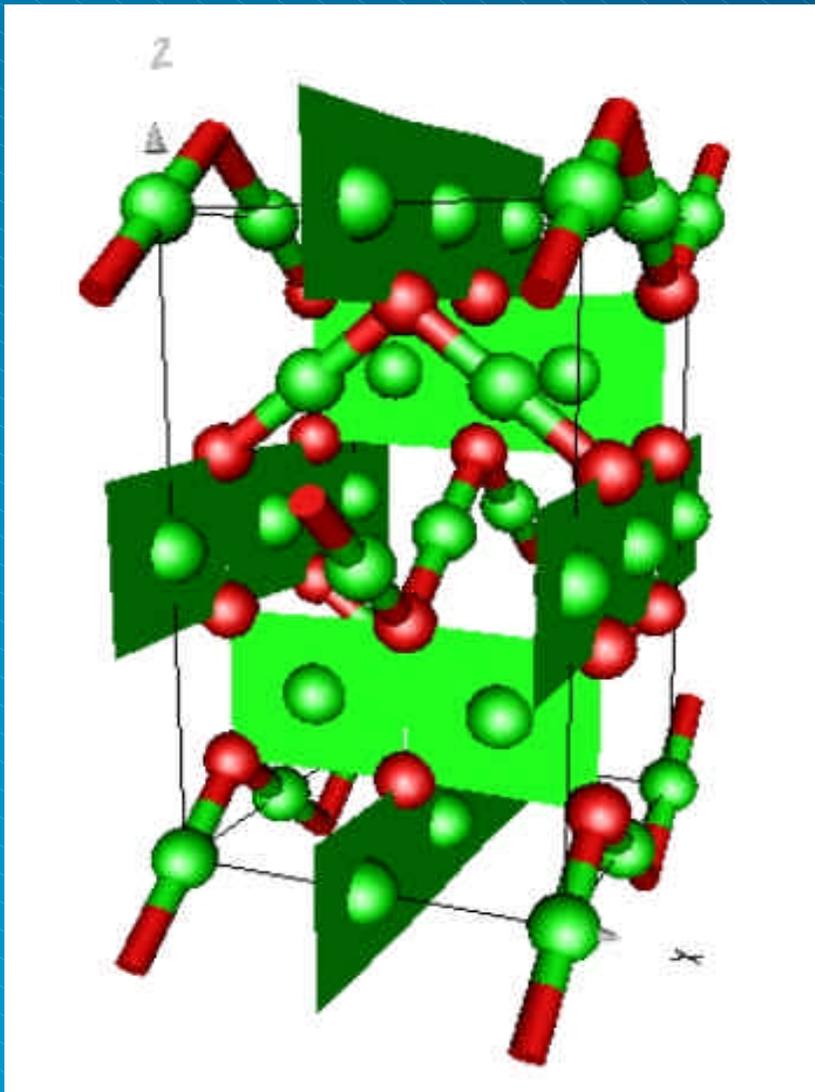
$\text{YBa}_2\text{Cu}_3\text{O}_7$  drawing from Capponi et al. Europhys Lett 3 1301 (1987)



# Valence Sum Calculations

What is the valence of Cu in  $\text{Cu}_4\text{O}_3$  ?

O'Keeffe, M. Boin, J. Am. Miner 63 180 (1978)



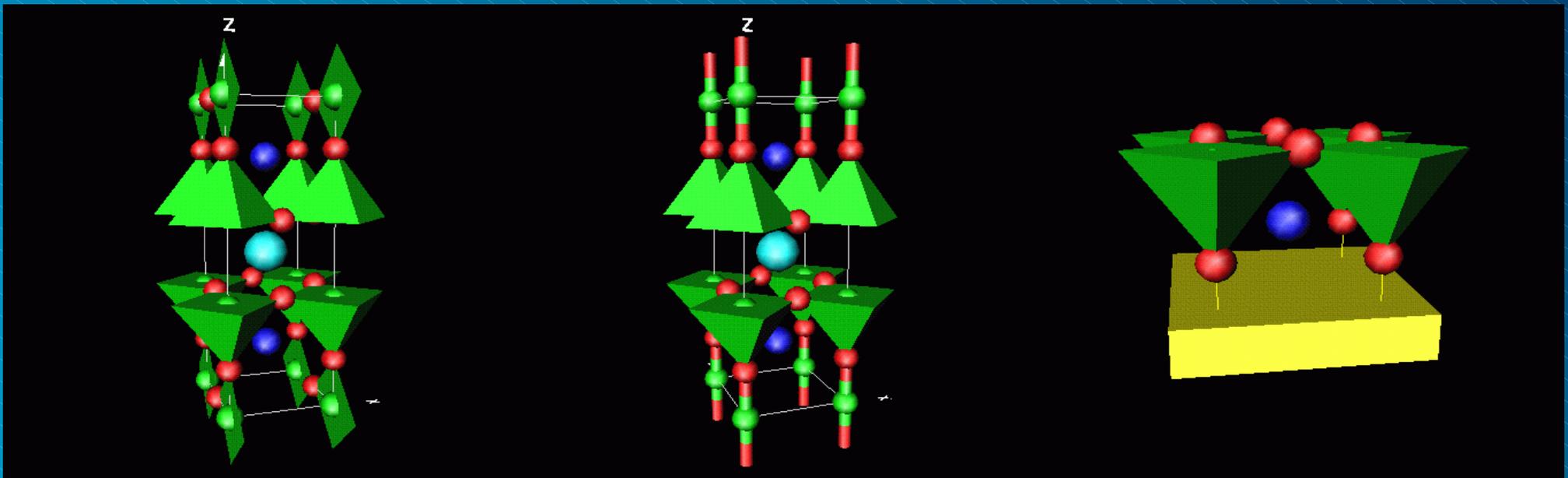
- Average Cu valence =  $2 \times 3/4 = 1.5$
- 2 types of Cu
  - $\text{Cu}^+$  at  $(0,0,0)$  with 2 oxygens
  - $\text{Cu}^{2+}$  at  $(0,0,1/2)$  with 4 oxygens
- Valence Sum  $V = \sum_i [\exp(\text{Ro} - \text{Ri}) / \text{B}]$ 
  - $\text{Ri}$  = Cu-O<sub>i</sub> bond lengths
  - $\text{Ro} = 1.610$  for  $\text{Cu}^+$  to  $\text{O}^{2-}$
  - $\text{B} = 0.370$
- Calculate  $\text{Ri}$  bond lengths & hence  $V$



# Valence Sums & "Charge Transfer"

Most cited neutron papers - "charge reservoir" concept in oxide superconductors

- Superc.  $\text{YBa}_2\text{Cu}_3\text{O}_7$
- Non-superc.  $\text{YBa}_2\text{Cu}_3\text{O}_6$
- Charge Reservoir



- [Cava, R. J. et al. \(1990\). Physica C. 165: 419 \(Bell labs/CNRS/ILL\)](#)
- Jorgensen, .D. et al. (1990) Phys. Rev. B41, 1863 (Argonne)



# Valence Sums & "Charge Transfer"

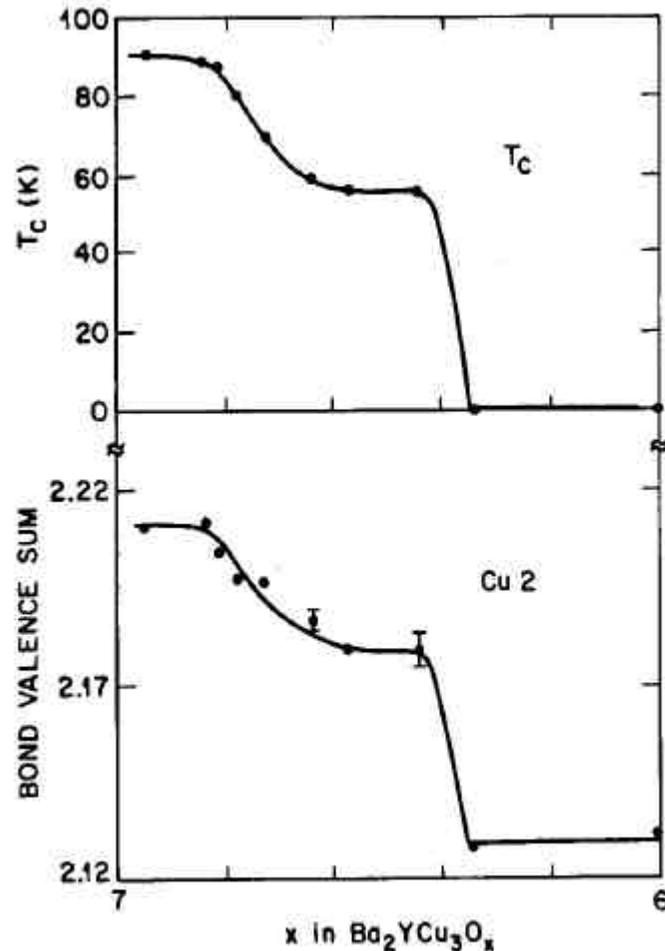


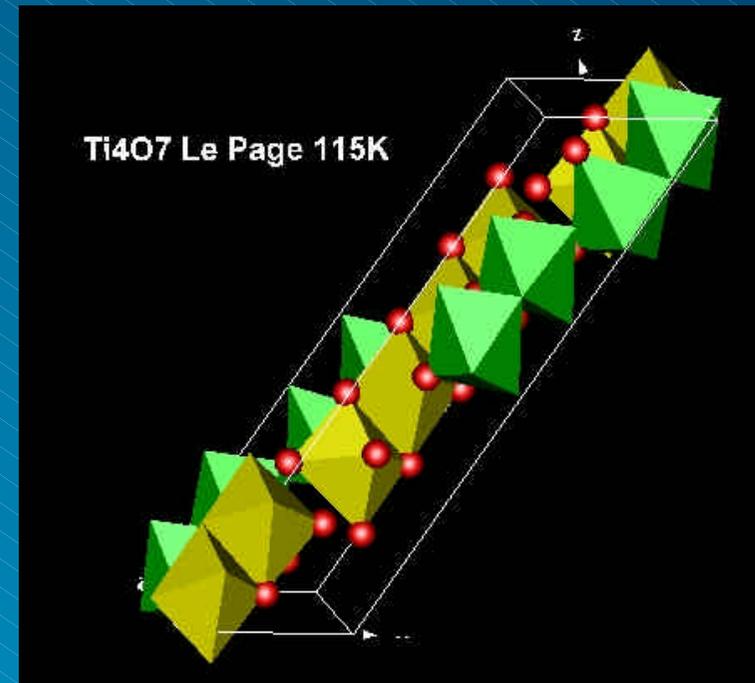
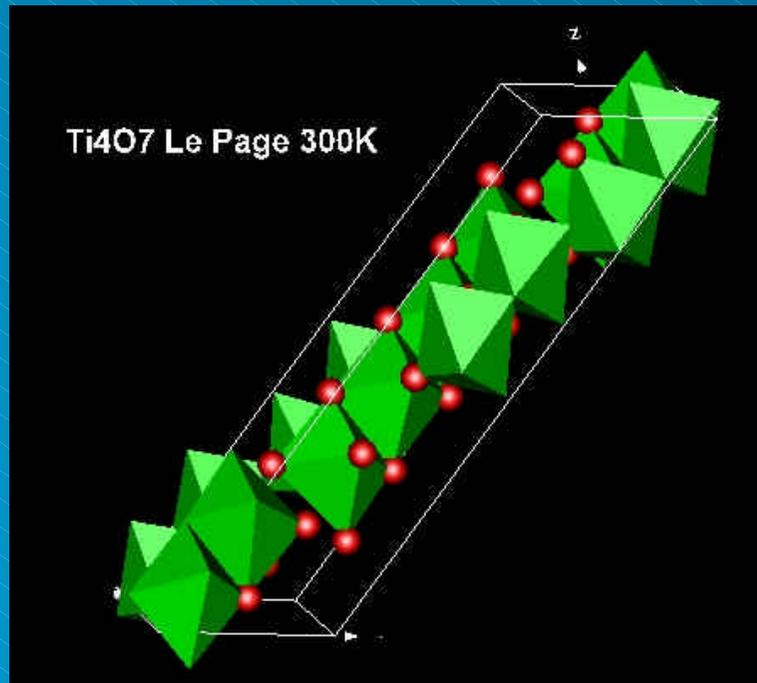
Fig. 16. Comparison of  $T_c$  and bond valence sum around the plane copper as a function of oxygen stoichiometry.

- Relation between bond lengths, charge transfer and superconducting  $T_c$
- The "Charge Reservoir" concept encouraged many chemists to successfully search for similar materials with different charge reservoir layers



# Electronic Order-Disorder

- Oxide superconductors, CMR, Vewey transition...
- Precise structural measurements vs temperature



- Example: charge ordering in  $\text{Ti}_4\text{O}_7$  (Le Page et al.)



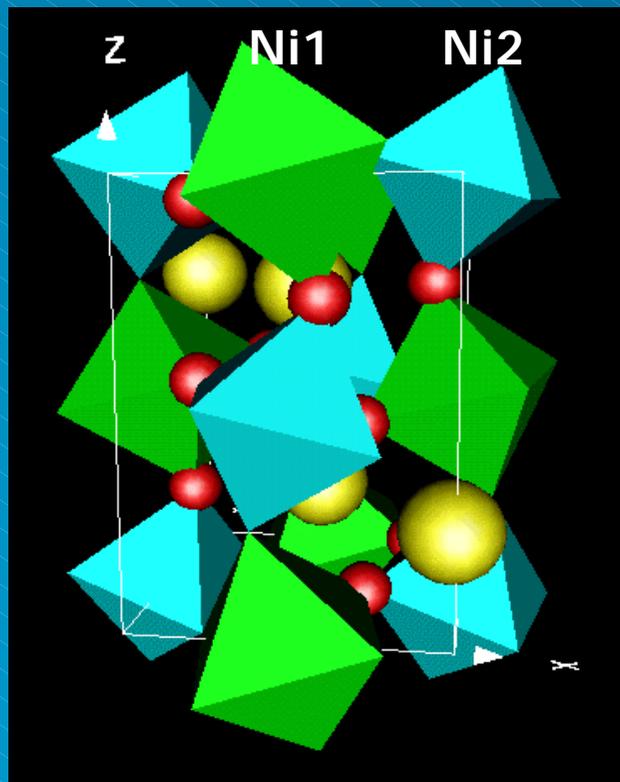
# Neutron Powder Diffraction

## Charge Transfer in $\text{YNiO}_3$

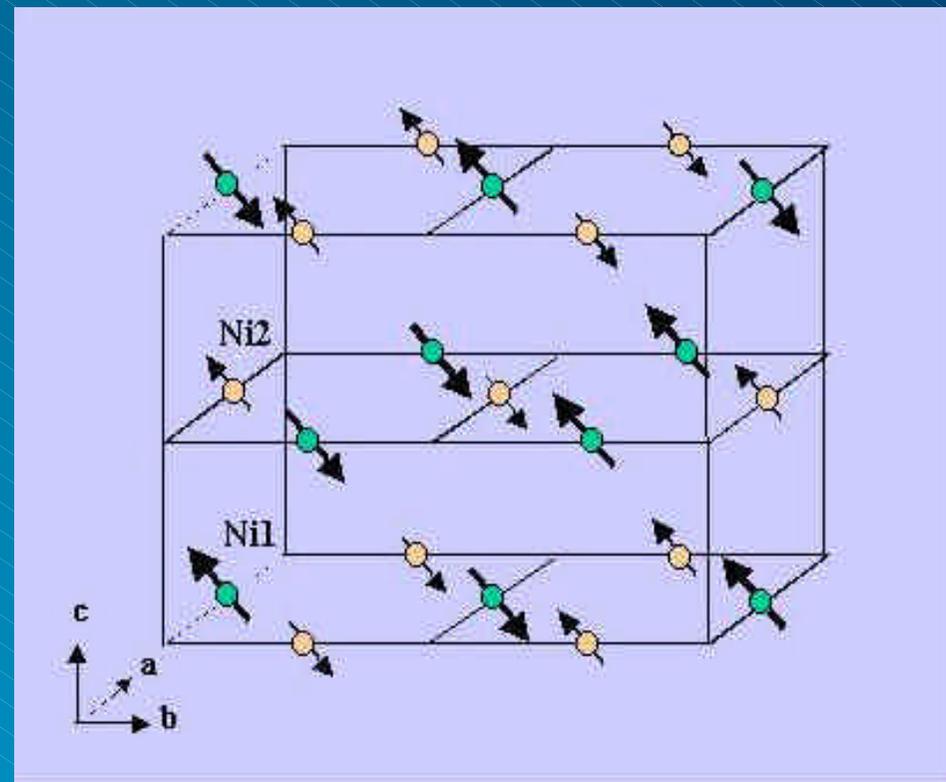
Marie-Theresa Fernandez-Diaz et al.

Combined ESRF, D1B and D2B data - Alonso J.A. et al (1999) PRL 82, 3873

Metallic Ortho.  $\text{YNiO}_3 \rightarrow$  Insulating Mono.  $\text{YNiO}_3$   $T < 582\text{K}$  Ni valence  $3-d$ ,  $3+d$



$V(\text{Ni1}) = 2.62$        $V(\text{Ni2}) = 3.17$



$M(\text{Ni1}) = -1.4 \mu_B$        $M(\text{Ni2}) = 0.7 \mu_B$

# Neutron Powder Diffraction

## Charge Transfer in $\text{YNiO}_3$

Marie-Theresa Fernandez-Diaz et al.

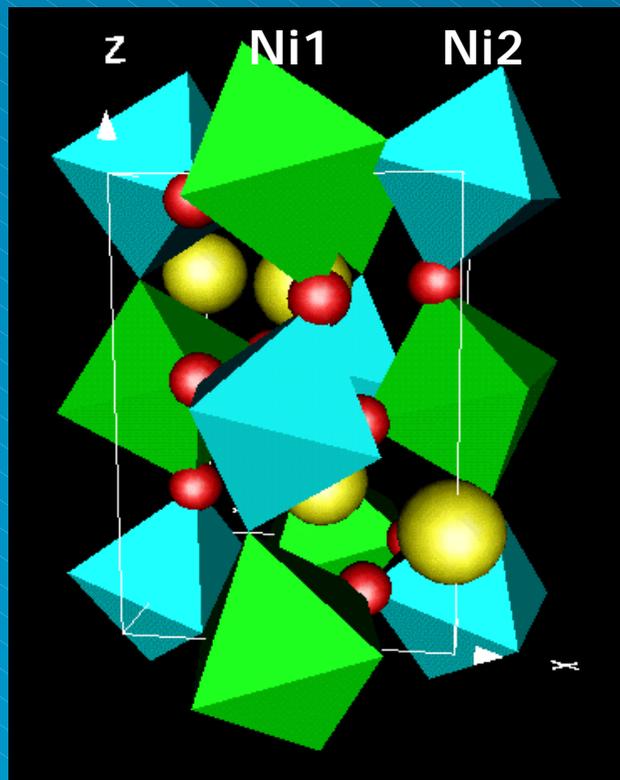
Alan Hewat



ILL Grenoble

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$V(\text{Ni1}) = 2.62$     $V(\text{Ni2}) = 3.17$

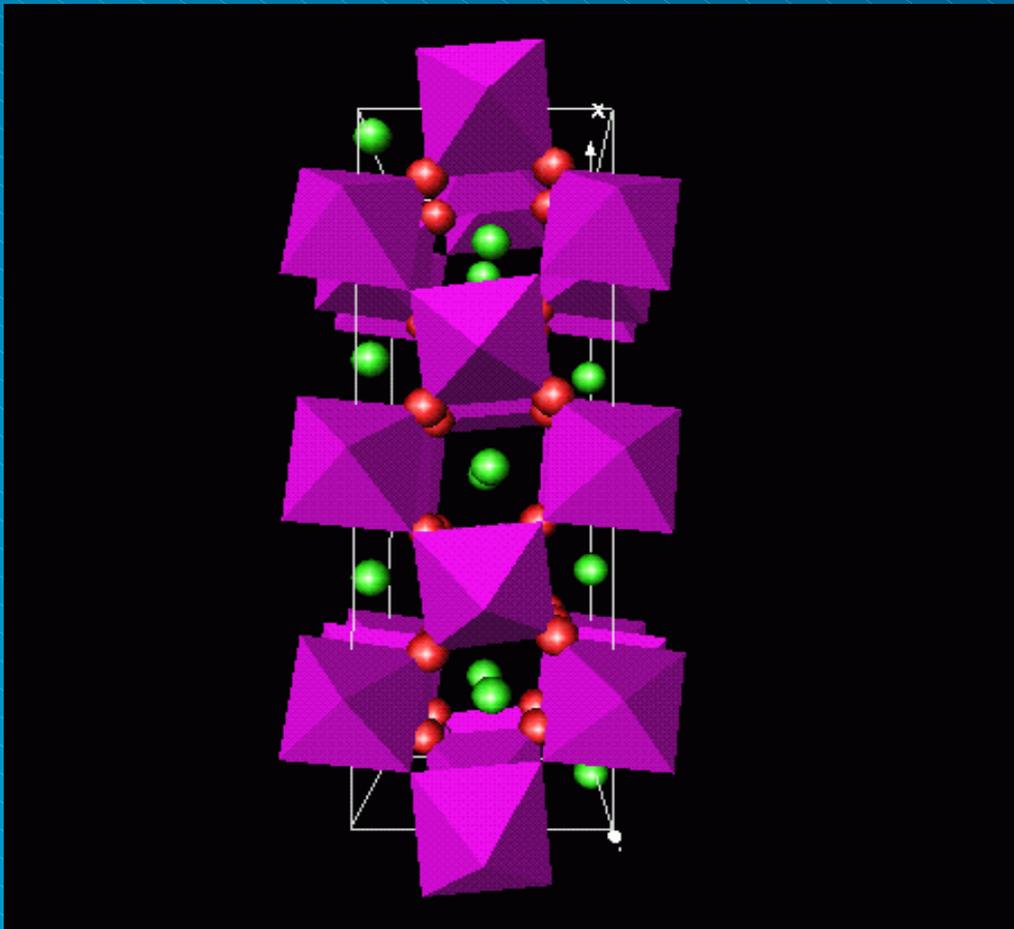
- Double evidence for charge transfer
  - Magnetic superstructure and different moments on Ni-sites
  - Different Ni-O distances around Ni1 and Ni2 sites mean 'charge transfer'
- Neutrons provide both. But need:
  - High resolution to resolve symmetry
  - High flux to see superstructure



# Giant Magneto-Resistive Ceramics



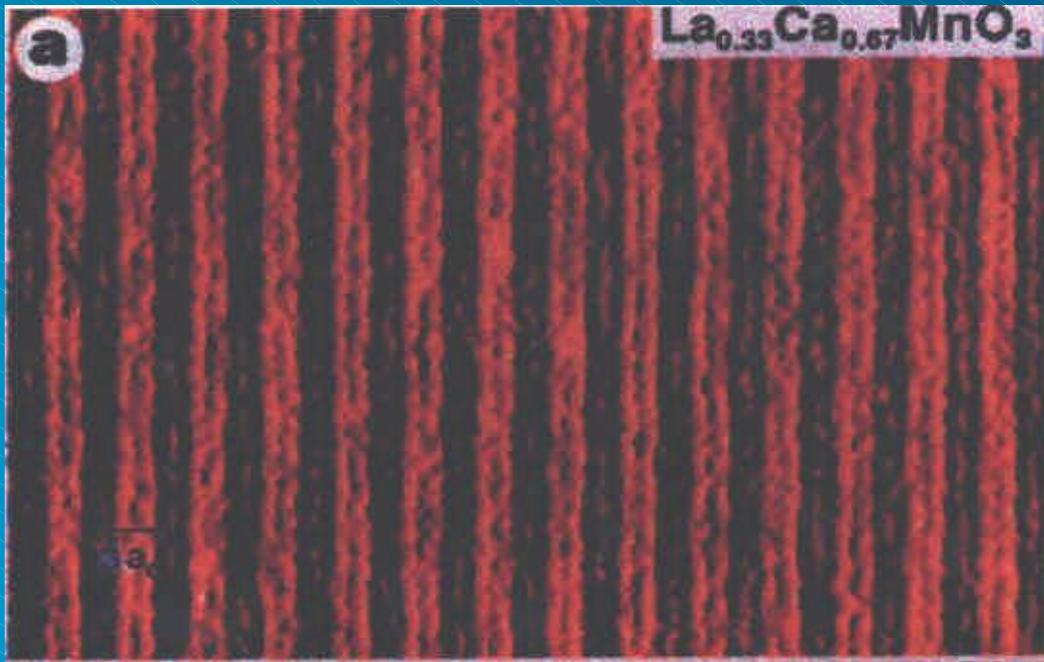
- Very large changes in electrical resistivity with temperature
- cf oxide superconductors
- mixed valence charge-ordering  $\text{Mn}^{3+}/\text{Mn}^{4+}$
- GMR effect near room temperature
- applications to magnetic storage of data (new high density IBM hard disks)





# GMR Stripes and Charge Ordering

1D-ordering ? Dimensionality important for theory.



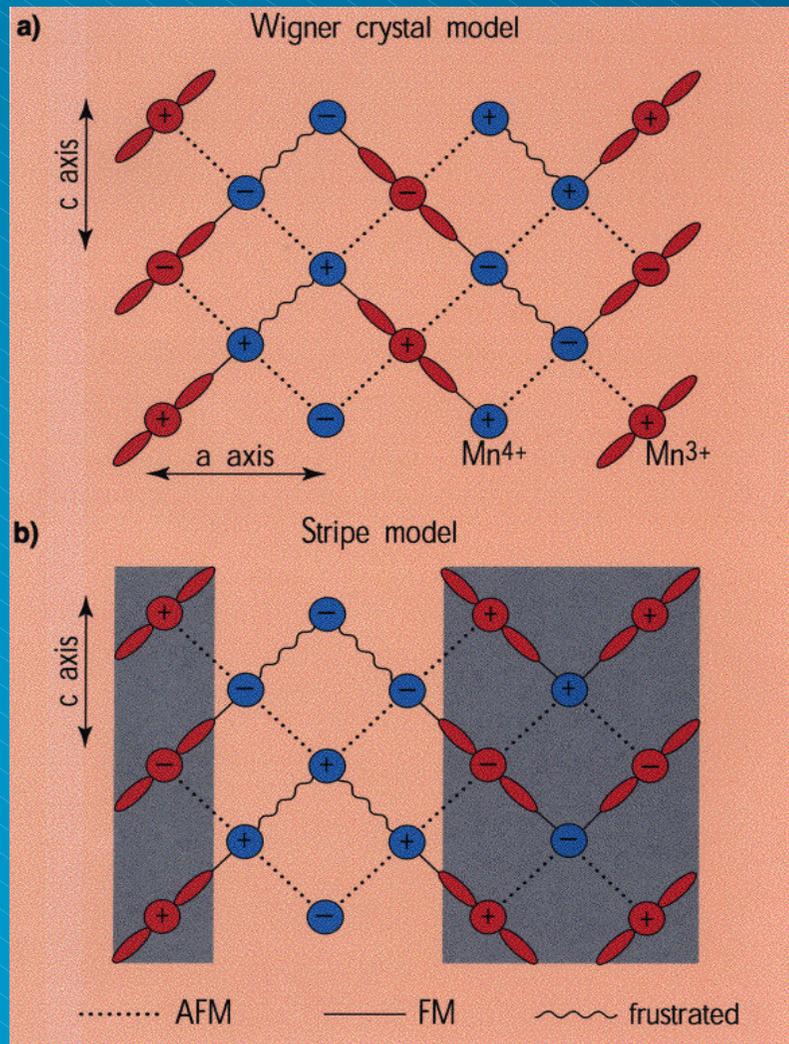
- Remarkable electron microscope images of 1D stripe pattern in GMR  $\text{La}_{0.33}\text{Ca}_{0.67}\text{MnO}_3$
- Evidence also for 1D ordering in high- $T_c$  superconductors ( $\text{Cu}^{3+}$  stripes, spin-ladders etc)

Mori et al. Nature (1998) 392,473  
Other papers in Phys. Rev. Letters



# GMR Stripes and Charge Ordering

1D-ordering ? Dimensionality important for theory.

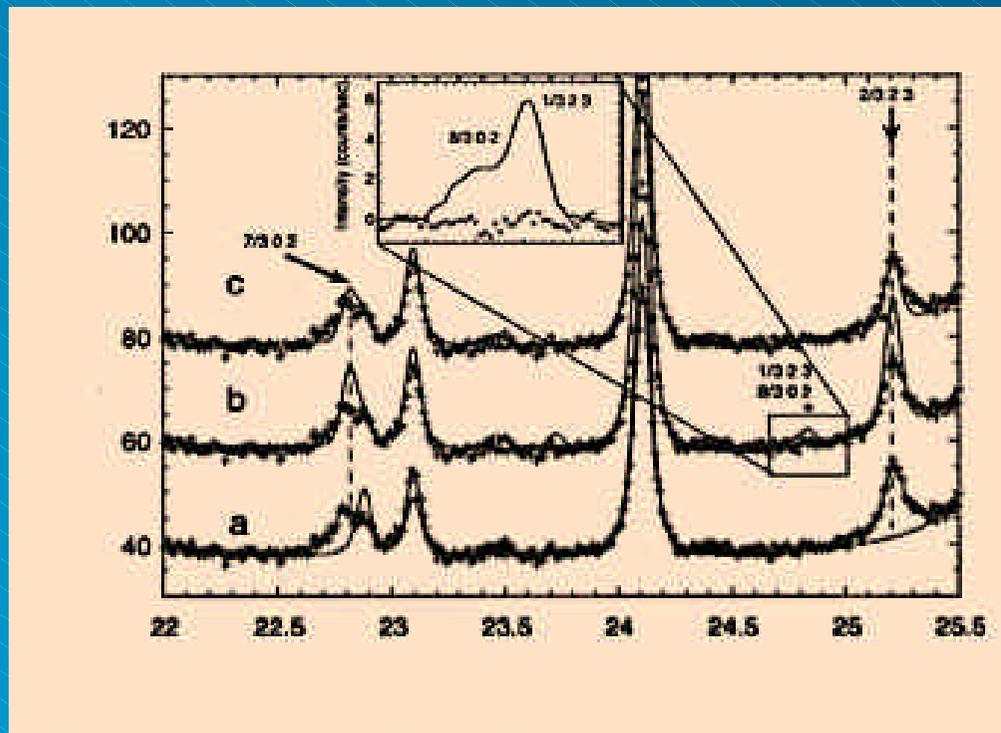


- Expect instead  $\text{Mn}^{3+}/\text{Mn}^{4+}$  to be uniformly distributed (2D Wigner crystal model of Goodenough)
- The 1D-stripe model would have very important consequences for the theory of superconductors and GMR oxides



# GMR Stripes and Charge Ordering

Neutron + Synchrotron Powder Diffraction



Radaelli et al. (1999) Phys. Rev B  
 X-ray work on X7A (BNL)  
 Neutron work on D2B (ILL)

- High resolution synchrotron powder data (Brookhaven) reveals true symmetry & ss
- High resolution neutron powder data (ILL Grenoble) allows refinement of real structure
  - a) Average Structure
  - b) Stripe Structure
  - c) Wigner Crystal Structure (best fit)
- The stripe structure is not supported



# Neutron Powder Diffraction

- What has been achieved ? Exciting new science ?
  - High impact even outside the crystallographic community
  - Magnetism, Superconductors, Giant Magneto-Resistance
- Why Neutrons ? Why not X-rays ?
  - Neutrons+X-rays complementary
  - Solution of structures with X-rays
  - Refinement of important details with neutrons - valence sums
- Why Powders ? Why not crystals ?
  - Crystals should be used when available
  - Much new work started with powders - high  $T_c$ , GMR...